# A New Type of Inorganic Polymer with Ordered SiSiGe Sequences

Polysilanes and polygermanes are a new class of inorganic polymers based on linear backbones of catenated silicon atoms and germanium atoms, respectively. These polymers show remarkable properties such as photoconductivity, 1-8 highly efficient photoluminescence, 4 conductivity induced by chemical doping,<sup>5</sup> and third-order nonlinear effects.<sup>6,7</sup> Recently, a theoretical approach suggested that Si/Ge copolymers with ordered sequences corresponded to one-dimensional superlattice structures.8 A one-dimensional superlattice would be of great scientific interest. However, a Wurtz type reaction of a mixture of organodichlorosilane and organodichlorogermane gives a random Si/Ge copolymer.<sup>9,10</sup> In this paper, we present a new idea that catenation of monomers with an SiGeSi sequence generates a periodic structure of SiSiGe sequences along a chain (Scheme I). This is the first report of the synthesis of a new type of inorganic polymer with ordered Si/Ge sequences.

The new polymer was synthesized in a three-step process. 11 First, 10g of sodium/potassium alloy (1:1 weight ratio) was dispersed in 300 mL of THF solution and refluxed. The mixture of 10.0 g (39 mmol) of di-n-butyldichlorogermane (Bu<sub>2</sub>GeCl<sub>2</sub>) and 16.9 g (150 mmol) of trimethylchlorosilane (Me<sub>3</sub>SiCl) was added dropwise to the solution. After 3 h of reflux, the reaction mixture was filtered and the filtrate was distilled under reduced pressure. A total of 2.43 g (7.3 mmol) of bis(trimethylsilyl)di-n-butylgermane ((Me<sub>3</sub>Si)<sub>2</sub>GeBu<sub>2</sub>) was obtained [yield: 18.9%, bp: 69 °C (0.3 mmHg)]. Second, 4 h of stirring of the mixture of 2.43 g of (Me<sub>3</sub>Si)<sub>2</sub>GeBu<sub>2</sub>, 10 g of Me<sub>3</sub>SiCl, and anhydrous aluminum chloride as a catalyst at 48 °C gave 530 mg (1.4 mmol) of bis(dimethylchlorosilyl)di-n-butylgermane ((Me<sub>2</sub>ClSi)<sub>2</sub>GeBu<sub>2</sub>) [yield: 19.2%, bp: 93 °C (0.5 mmHg)]. Third, 100 mg of sodium was dispersed in 10 mL of refluxed toluene. A total of 530 mg of (Me<sub>2</sub>ClSi)<sub>2</sub>GeBu<sub>2</sub> was added dropwise to the suspension and stirred for 3 h. The reaction mixture was filtered, and the filtrate was poured dropwise into excess ethanol. A total of 110 mg of polymer 1 was obtained (overall yield based on Bu<sub>2</sub>GeCl<sub>2</sub>: 0.94%).

For the characterization of polymer 1, two random copolymers consisting of Si and Ge were also prepared by the ordinary method (Scheme II). The mixture of 7.26 g (39 mmol) of 1,1,2,2-tetramethyl-1,2-dichlorodisilane (ClMe<sub>2</sub>SiSiMe<sub>2</sub>Cl) and 10.0 g (39 mmol) of Bu<sub>2</sub>GeCl<sub>2</sub> and added dropwise to 150 mL of toluene including 4.3 g (197 mmol) of sodium dispersion. After 6 h of reflux, the reaction mixture was filtered and the filtrate was poured dropwise into excess ethanol. A total of 384 mg of poly-(tetramethyldisilane-ran-di-n-butylgermane) (polymer 2) was obtained. Poly(dimethylsilane-ran-di-n-butylgermane) (polymer 3) was synthesized by a similar method using 10 g (77 mmol) of dimethyldichlorosilane (Me<sub>2</sub>SiCl<sub>2</sub>), 10 g of Bu<sub>2</sub>GeCl<sub>2</sub>, and 6.0 g of sodium dispersion. A total of 390 mg of polymer 3 was obtained.

The resulting polymers were characterized by GPC, X-ray fluorescence analysis (XFA),  $^{12}$  UV absorption,  $^{1}$ H NMR (200 MHz), and  $^{29}$ Si NMR (39 MHz). All NMR spectra were measured in tetrahydrofuran- $d_8$  at 20 °C. Table I shows the results of these analyses, and Figure 1 shows the  $^{29}$ Si NMR spectra. The atomic ratios (Si/Ge) of polymers 2 and 3 are quite different from the nominal compositions. In contrast, the atomic ratio (Si/Ge) of polymer 1 is 2.00, as expected from the procedure.

## Scheme I

### Formation of Random Si/Ge Structure

#### Formation of Ordered Si/Ge Structure

$$\begin{array}{c} \text{R}_{1} \ \ R_{2} \ \ R_{1} \\ \text{R}_{1} \ \ R_{2} \ \ R_{1} \\ \text{R}_{1} \ \ R_{2} \ \ R_{1} \\ \end{array} \qquad \begin{array}{c} \begin{array}{c} \begin{array}{c} R_{1} \ \ R_{2} \ \ R_{1} \\ \text{Si-Ge-Si-Si-Ge-Si-Ni-Me-Bu-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Ge-Ni-Me-Ni-Me-Ni-Ge-Ni-M$$

In the <sup>29</sup>Si NMR spectra, polymer 2 has two major multiple peaks near -38 and -29 ppm and polymer 3 has three near -37, -29, and -20 ppm. Two triad configurational sequences are predicted for polymer 2 (SiSi\*Si and SiSi\*Ge) and three for copolymer 3 (SiSi\*Si, SiSi\*Ge, and GeSi\*Ge). The results of <sup>29</sup>Si NMR are consistent with this prediction. Since <sup>29</sup>Si NMR chemical shifts for catenated silicon atoms with methyl substituents can be observed in the range of -36 to -40 ppm,13 the peaks observed near -36 ppm in polymers 2 and 3 are assigned to Si\* with a SiSi\*Si triad configurational sequence. The peaks near -29 ppm and near -20 ppm are assigned to Si\* with SiSi\*Ge and GeSi\*Ge triad configurational sequences, respectively, because a germanium atom causes the signals of a nearby silicon atom to be shifted downfield. The <sup>1</sup>H NMR spectra of polymers 2 and 3 also reflect the number of triad configurational sequences. Polymer 2 has two peaks that are assigned to SiCH\*3, and polymer 3 has three. In addition, the multiplicity of the <sup>29</sup>Si peaks shown in polymers 2 and 3 is attributed to several pentad configurational sequences such as SiGeSi\*SiGe and SiGeSi\*SiSi. These results mean that polymers 2 and 3 have random Si/Ge chain structures.

polymer	nominal composn of Si/Ge (atomic ratio)	yields of polymer, %	mol wts (weight-averaged) <sup>c</sup>	atomic ratio of Si/Ge <sup>d</sup>	λ <sub>max</sub> , e nm	<sup>1</sup> H NMR chemical shifts of SiCH* <sub>3</sub> , ppm from TMS
1	$2.0^{b}$	25.6	6 000	2.00	298.6	0.391
2	2.0	3.2	11 000	1.16	308.5	0.278 (1.0), 0.396 (0.8)
3	2.0	7.9	13 000	0.35	312.0	0.263 (1.0), 0.385 (0.8), 0.487 (0.8)

<sup>a</sup> All polymers were purified twice by reprecipitation with a toluene—ethanol system. Toluene-soluble fractions of polymers 2 and 3 were used in measurements. <sup>b</sup> Atomic ratio of the monomer. <sup>c</sup> Molecular weights were determined by GPC based on a polystyrene standard. <sup>d</sup> Compositional ratios were determined by XFA analysis. <sup>e</sup> UV spectra were measured in hexane at 25 °C. <sup>f</sup> TMS: tetramethylsilane. Values in parentheses are relative intensities.

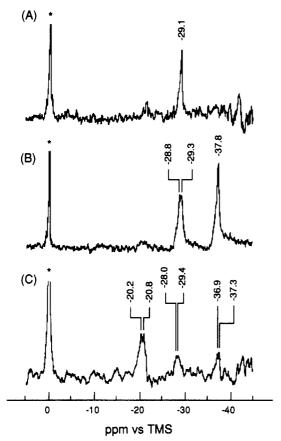


Figure 1. <sup>29</sup>Si NMR spectra (39 MHz) in tetrahydrofuran- $d_8$  at 20 °C of (a) polymer 1, (b) polymer 2, and (c) polymer 3. Asterisks indicate the tetramethylsilane internal signals.

In contrast, polymer 1 has one peak assigned to SiCH\*<sub>3</sub> at 0.391 ppm in the <sup>1</sup>H NMR spectrum and only one peak at -29.1 ppm in the <sup>29</sup>Si NMR spectrum. This peak in the <sup>29</sup>Si NMR is assigned to Si\* with an SiSi\*Ge triad configurational sequence because of the similar chemical shift to the peaks observed in polymers 2 and 3. The singularity of the <sup>29</sup>Si signal means that polymer 1 has only one type of pentad configurational sequence. There-

fore, polymer 1 consists of ordered SiSiGe sequences.

In summary, the present study shows that polymer 1 is in a periodic SiSiGe one-dimensional structure. Polymer 1 is the first example of a soluble one-dimensional superlattice. We are currently measuring the properties of these polymers, and they will be reported elsewhere.

Acknowledgment. We thank Dr. Masao Morita for his helpful discussion on NMR study and Hiroshi Kojima for his assistance in measuring molecular weight distributions.

#### References and Notes

- Kepler, R. G.; Zeigler, J. M.; Harrah, L. A.; Kurtz, S. R. Phys. Rev. B 1987, 35, 2818.
- (2) Fujino, M. Chem. Phys. Lett. 1987, 136, 451.
- (3) Abkowitz, M.; Stolka, M. J. Non-Cryst. Solids 1989, 114, 342.
- (4) Miller, R. D.; Michl, J. Chem. Rev. 1989, 89, 1359.
- (5) West, R. J. Organomet. Chem. 1986, 300, 327.
- (6) Kajzar, F.; Messier, J.; Rosilio, C. J. Appl. Phys. 1986, 60, 3040.
  (7) Baumert, J. C.; Bjorklund, G. C.; Jundt, D. H.; Jurich, M. C.; Looser, H.; Miller, R. D.; Rabolt, J.; Sooriyakumaran, R.; Swalen, J. D.; Twieg, R. J. Appl. Phys. Lett. 1988, 53, 1147.
- (8) Takeda, K.; Shiraishi, K.; Matsumoto, N. J. Am. Chem. Soc. 1990, 112, 5043.
- (9) Trefonas, P.; West, R. J. Polym. Chem., Polym. Chem. Ed. 1985, 23, 2099.
- (10) Miller, R. D.; Sooriyakumaran, R. J. Polym. Sci., Part A: Polym. Chem. 1987, 25, 111.
- (11) This procedure is a modification of a previously reported method: Fujino, M.; Matsumoto, N.; Ban, H.; Sukegawa, K. J. Polym. Sci., Part C: Polym. Lett. 1988, 26, 109.
- (12) Samples and a reference standard (Si/Ge = 1/1 weight ratio) were prepared by casting on acrylic resin substrates. Characteristic X-rays used in the analysis are Ge Kα, Ge Lβ, and Si Kα. Si/Ge atomic ratios were converted from intensity ratios of Ge Kα/Si Kα and Ge Lβ/Si Kα.
- (13) Ishikawa, M.; Iyoda, J.; Ikeda, H.; Kotake, K.; Hashimoto, T.; Kumada, M. J. Am. Chem. Soc. 1981, 103, 4845.

# Hiroaki Isaka,\* Michiya Fujiki, Masaie Fujino, and Nobuo Matsumoto

Materials Science Research Laboratory NTT Basic Research Laboratories 3-9-11, Midoricho, Musashino, Tokyo 180, Japan

> Received December 4, 1990 Revised Manuscript Received March 4, 1991